



Transient Thermal and Electrical Co-Optimization of BEOL Top-Gated ALD In_2O_3 FETs Toward Monolithic 3-D Integration

Pai-Ying Liao¹, Dongqi Zheng, Sami Alajlouni², Zhuocheng Zhang³, Mengwei Si¹, *Member, IEEE*, Jie Zhang¹, *Member, IEEE*, Jian-Yu Lin, Tatyana I. Feygelson, Marko J. Tadjer⁴, *Senior Member, IEEE*, Ali Shakouri, and Peide D. Ye¹, *Fellow, IEEE*

Abstract—In this work, the transient thermal and electrical characteristics of top-gated (TG), ultrathin, atomic-layer-deposited (ALD), back-end-of-line (BEOL) compatible indium oxide (In_2O_3) transistors on various thermally conductive substrates are co-optimized by visualization of the self-heating effect (SHE) utilizing an ultrafast high-resolution (HR) thermo-reflectance (TR) imaging system and overcome the thermal challenges through substrate thermal management and short-pulse measurement. At the steady-state, the temperature increase (ΔT) of the devices on highly resistive silicon (HR Si) and diamond substrates are roughly 6 and 13 times lower than that on a SiO_2/Si substrate, due to the much higher thermal conductivities (κ) of HR Si and diamond. Consequently, the ultrahigh drain current (I_D) of 3.7 $\text{mA}/\mu\text{m}$ at drain voltage (V_{DS}) of 1.4 V with direct current (dc) measurement is achieved with TG ALD In_2O_3 FETs on a diamond substrate. Furthermore, transient thermal study shows that it takes roughly 350 and 300 ns for the devices to heat-up and cool-down to the steady-states, being independent of the substrate. The extracted thermal time constants of heat-up (τ_h) and cool-down (τ_c) processes are 137 and 109 ns, respectively. By employing electrical short-pulse measurement with a pulsewidth (t_{pulse}) shorter than τ_h , the SHE

can be significantly reduced. Accordingly, a higher I_D of 4.3 $\text{mA}/\mu\text{m}$ is realized with a 1.9-nm-thick In_2O_3 FET on HR Si substrate after co-optimization. Besides, to integrate BEOL-compatible ALD In_2O_3 transistors on the front-end-of-line (FEOL) devices with the maintenance of the satisfactory heat dissipation capability, a FEOL-interlayer-BEOL structure is proposed where the interlayer not only electrically isolates the FEOL and BEOL devices but also serves as a thermally conductive layer to alleviate the SHE.

Index Terms—Atomic layer deposition, indium oxide (In_2O_3), self-heating effect (SHE), thermal engineering, thermo-reflectance (TR).

I. INTRODUCTION

DESPITE the wide application in the display industry [1], oxide semiconductors attract more interest recently as back-end-of-line (BEOL)-compatible channel materials for thin-film transistor (TFT) applications even toward monolithic 3-D (M3D) integration [2], [3], [4], [5], [6], [7], [8], [9], [10], [11], [12], [13], [14], [15], [16], [17]. Among them, atomic-layer-deposited (ALD) indium oxide (In_2O_3) is of great interest due to its outstanding properties such as wafer-scale uniformity, high carrier mobility, BEOL compatibility, ambient stability, and capability of conducting ultrahigh current. Especially, the ALD growth method provides not only the angstrom-scale thickness control but also the atomically smooth surface conformality on 3-D surface structures such as deep trenches and side walls [2], [3], [4], [5], [6], [7], [8]. It has been demonstrated that back-gated (BG) ALD In_2O_3 transistors can be realized with an ultra-thin body of 0.5 nm, which is only a couple of atoms thick [6]. Besides, a high drain current (I_D) up to 2.2 $\text{mA}/\mu\text{m}$ at drain bias (V_{DS}) of 0.7 V is reached by 1.5-nm-thick BG In_2O_3 devices operated at enhancement mode (E-mode) [14].

Nevertheless, even though top-gated (TG) devices are particularly desired for practical applications, the explorations of In_2O_3 FETs mostly focus on BG structure. This is mainly due to the two challenges of defect induction during the formation of the high- k TG dielectric and severe self-heating

Manuscript received 16 November 2022; revised 31 December 2022; accepted 4 January 2023. Date of publication 16 January 2023; date of current version 24 March 2023. This work was supported in part by the Semiconductor Research Corporation (SRC), Nanoelectronic Computing Research (nCORE) Innovative Materials and Processes for Accelerated Compute Technologies (IMPACT) Center and in part by the Applications and Systems driven Center for Energy-Efficient Integrated NanoTechnologies (ASCENT) Center through the SRC/Defense Advanced Research Projects Agency (DARPA) Joint University Microelectronics Program (JUMP). This article is an extended version of a paper presented at IEDM 2022. The review of this article was arranged by Editor D. Triyoso. (Corresponding author: Peide D. Ye.)

Pai-Ying Liao, Dongqi Zheng, Sami Alajlouni, Zhuocheng Zhang, Mengwei Si, Jie Zhang, Jian-Yu Lin, Ali Shakouri, and Peide D. Ye are with the School of Electrical and Computer Engineering and the Birk Nanotechnology Center, Purdue University, West Lafayette, IN 47907 USA (e-mail: yep@purdue.edu).

Tatyana I. Feygelson and Marko J. Tadjer are with the Electronics Science and Technology Division, United States Naval Research Laboratory, Washington, DC 20375 USA.

Color versions of one or more figures in this article are available at <https://doi.org/10.1109/TED.2023.3235313>.

Digital Object Identifier 10.1109/TED.2023.3235313

effect (SHE) with high power density (PD) [7], [8]. The root cause of the former issue is that the formation of the high- k dielectric stack, ALD HfO_2 , likely fetches oxygen not only from the oxygen precursor but also the In_2O_3 channel underneath, which induces oxygen vacancies at the channel. Fortunately, this can be partly resolved by lowering the growth temperature of ALD HfO_2 from 200 °C down to 120 °C, followed by rapid thermal annealing (RTA) treatment in the O_2 environment [5], [8]. On the other hand, the latter challenge remains a bottleneck of TG ALD In_2O_3 devices.

In this work, an ultrafast high-resolution (HR) thermo-reflectance (TR) imaging system is introduced to visualize the transient and steady-state characteristic ΔT of TG In_2O_3 devices on different substrates to address the thermal issues. The substrate substitution method [7] is first applied to alleviate SHE. At the steady states, the SHE is mitigated by a factor of 6 and 13 with HR Si and diamond substrate, respectively, compared to SiO_2/Si substrate. Next, in order to understand the transient thermodynamics of the self-heating process of TG ALD In_2O_3 transistors, transient TR measurement is performed. The extracted time constants τ_h and τ_c are roughly 137 and 109 ns, respectively. By resolving SHE, an ultrahigh I_D of 3.7 $\text{mA}/\mu\text{m}$ is realized with 2.5-nm-thick In_2O_3 devices on the diamond substrate under direct current (dc) measurement. Furthermore, by pulse measurement with pulse widths shorter than τ_h , even higher I_D of 4.3 $\text{mA}/\mu\text{m}$ is achieved with 1.9-nm-thick In_2O_3 transistors on HR Si substrate under pulse measurement. Besides, BEOL-compatible TG ALD In_2O_3 transistors are built on top of front-end-of-line (FEOL) Si devices with a thick plasma-enhanced chemical-vapor-deposited (PECVD) silicon nitride (SiN_x) in between, showing potentials of combining FEOL devices with BEOL devices together toward M3D integration. This article is an extended version of an accepted 2022 IEEE International Electron Devices Meeting (IEDM 2022) paper with more details and FEOL-BEOL integration included.

II. DEVICE FABRICATION AND PERFORMANCE

Fig. 1(a) exhibits the schematic of TG ALD In_2O_3 transistors. Different κ -value substrates including SiO_2/Si , sapphire, HR Si, and diamond from 1.5 to 2200 $\text{W}\cdot\text{m}^{-1}\cdot\text{K}^{-1}$, are used. After solvent cleaning, a 1.6–2.5 nm In_2O_3 layer was conformally grown by ALD at 225 °C with trimethylindium (TMI) and H_2O as the In and O precursors, respectively, followed by an Ar/BCl_3 dry-etching step for device isolation. Then, 45 nm Ni was deposited as source/drain (S/D) contacts by e-beam lithography (EBL), e-beam evaporation, and a lift-off step. Next, a 7 nm HfO_2 top dielectric layer was formed by ALD at 120 °C with $[(\text{CH}_3)_2\text{N}]_4\text{Hf}$ (TDMAHf) and H_2O as the Hf and O precursors, respectively. Finally, the top-gate metal of 30/20 nm Au/Ni was deposited with the same process as the S/D metal, followed by an RTA treatment at 250 °C–300 °C in an O_2 environment for 2–4 min. The overall thermal budget is as low as 300 °C, making it BEOL-compatible.

Fig. 1(b) and (c) presents the electrical characteristics of a TG ALD In_2O_3 FET with a channel length (L_{ch}) of 600 nm and a thin channel thickness (T_{ch}) of 1.6 nm on a SiO_2/Si sub-

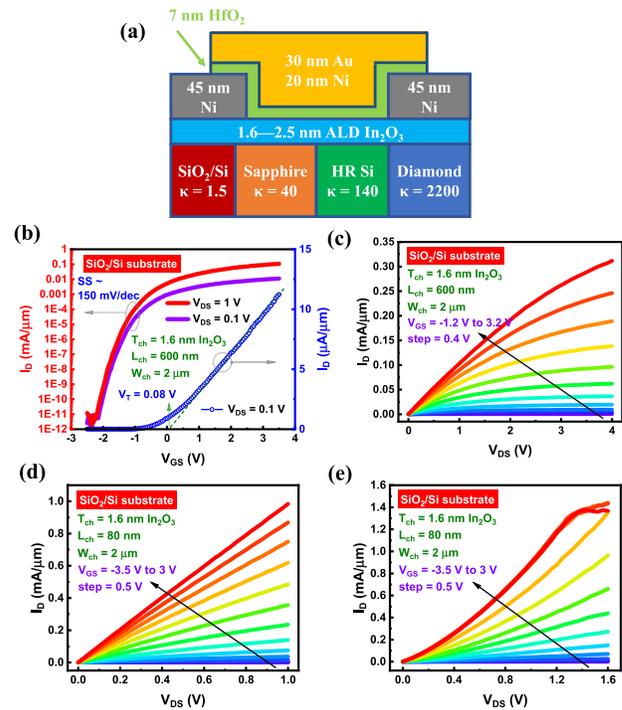


Fig. 1. (a) Device schematic of a TG ALD In_2O_3 FET with various thermally conductive substrates. The unit of thermal conductivity (κ) is $\text{W}\cdot\text{m}^{-1}\cdot\text{K}^{-1}$. (b) Transfer. (c) Output characteristics of a TG ALD In_2O_3 FET with long L_{ch} of 600 nm and thin T_{ch} of 1.6 nm on a SiO_2/Si substrate operated at enhancement-mode (E-mode). ON-OFF ratio of 10 orders of magnitude and SS value of 150 mV/dec are performed. (d) Output characteristics of a TG ALD In_2O_3 FET with short L_{ch} of 80 nm on a SiO_2/Si substrate. (e) Severe SHE deteriorates the device performance of a TG ALD In_2O_3 FET with high PD.

strate operated at enhancement-mode. The extracted threshold voltage (V_T) and subthreshold swing (SS) are 0.08 V and 150 mV/dec, respectively. ON/OFF ratio of 10 orders of magnitude is obtained due to the wide bandgap of In_2O_3 , indicating great switching behaviors. Fig. 1(d) shows the I_D - V_{DS} curves of a similar device with a shorter L_{ch} of 80 nm, exhibiting a maximum I_D of 1 $\text{mA}/\mu\text{m}$ at V_{DS} of 1 V which is much higher than the long channel devices. However, as the applied V_{DS} increases to 1.6 V as shown in Fig. 1(e), serious SHE happens due to low- κ of SiO_2 , and the device becomes unstable. The curves perform Schottky-like behaviors, and the current gradually deteriorates with higher and higher PD. This is the thermal bottleneck for high I_D In_2O_3 FETs, and therefore, the ultrafast HR TR imaging system is introduced in the following sections to visualize and resolve the thermal challenges.

III. STEADY-STATE CHARACTERIZATION

The ultrafast HR TR imaging equipment employed to systematically investigate and resolve the SHE is illustrated in Fig. 2(a). During the measurement, the device under test is applied by periodic V_{DS} pulses and a constant V_{GS} bias, where the source is common, to induce the SHE. A high-speed green light LED (wavelength of 530 nm) is equipped to illuminate the device surface for the synchronized charge-coupled device (CCD) camera to capture the surface reflectance. Fig. 2(b) demonstrates its working mechanism. As V_{DS} pulses start,

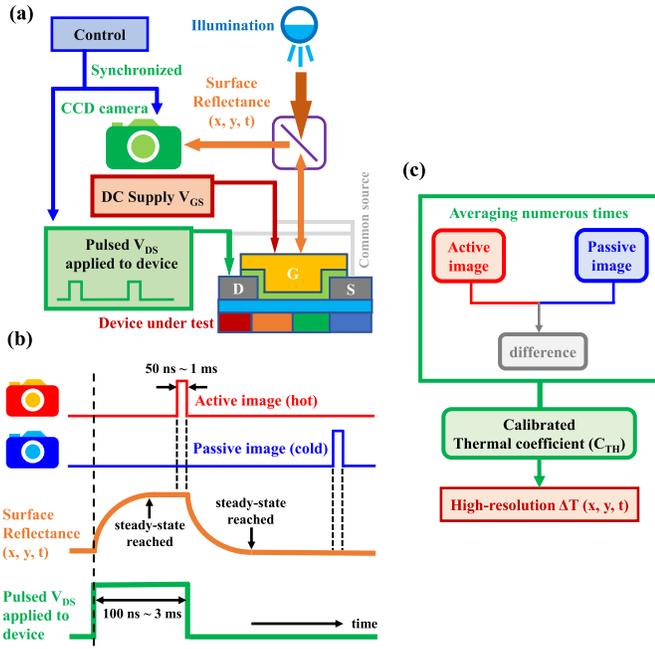


Fig. 2. (a) Schematic illustration of the ultrafast HR TR imaging system setup. (b) Working mechanism of the ultrafast HR TR imaging equipment in time domain. (c) Transformation from TR signal to a temperature scale.

the device is turned on and accordingly self-heats up, and TR signals are captured as an active image after the steady-state is reached as shown in red. Similarly, when V_{DS} pulses end, the device is turned off and hence naturally cools down to ambient temperature, and TR signals are captured as a passive image after the steady-state is reached as shown in blue. This process is repeated numerous times, and the difference between the active and passive images is averaged consequently and transformed into a temperature scale by dividing by the calibrated thermal coefficient of the surface material ($C_{TH} = -2.5 \times 10^{-4} \text{ K}^{-1}$ [18], [19], [20]) to obtain the final HR ΔT distribution image at the steady-state as shown in Fig. 2(c).

Fig. 3 presents the observed ΔT distribution of the TG In_2O_3 transistors around the channel region. All the devices under test have identical structures and dimensions of 400 nm L_{ch} and 2 μm W_{ch} , and the only difference is the substrate in use. Clearly, the one on a SiO_2/Si substrate is the most self-heated while the one on a diamond substrate is the least. The cross sections of the ΔT along the channel width (W_{ch}) direction normalized by PD is demonstrated in Fig. 4(a) where the PD is calculated by $(I_D \times V_{DS}) / (L_{ch} \times W_{ch})$. Similar bell-like shapes are observed in all the cases. The maximum ΔT of the devices with individual substrates and different PD is plotted in Fig. 4(b) where a linear relationship is obtained despite the substrate. The inversed slopes of the regression lines indicate the capability of the substrate to dissipate the generated Joule heat in the channel. To compare the thermal dissipation capabilities of the substrates, the ΔT is extrapolated to a constant PD of 16 kW/mm^2 in each case as shown in Fig. 4(c). Fig. 4(c) implies that the higher κ of the substrate is, the lower the maximum normalized ΔT will be. Sapphire ($\kappa = 40 \text{ W} \cdot \text{m}^{-1} \cdot \text{K}^{-1}$ [21]), HR Si ($\kappa = 140 \text{ W} \cdot \text{m}^{-1} \cdot \text{K}^{-1}$ [22]),

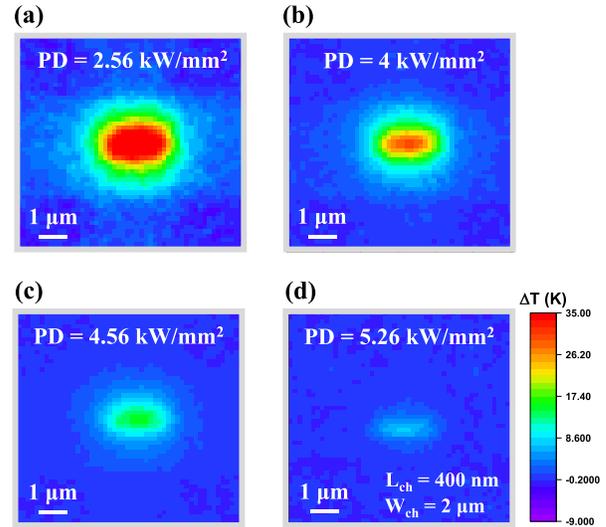


Fig. 3. SHE visualization of TG In_2O_3 FETs in experiments with substrates of (a) SiO_2/Si , (b) sapphire, (c) highly resistive Si, and (d) diamond with various PD. The L_{ch} and W_{ch} are 400 nm and 2 μm , respectively, for all the devices. The device on a SiO_2/Si substrate is the most self-heated while the one on a diamond substrate is the least.

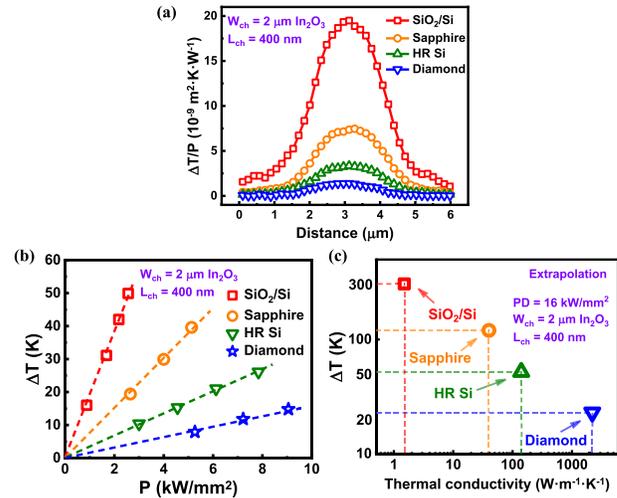


Fig. 4. (a) Cross sections of PD-normalized ΔT of TG ALD In_2O_3 FETs with different substrates. (b) ΔT extraction of TG ALD In_2O_3 devices with different substrates and power densities. Linear relationship is agreed in all the cases. (c) Comparison of the extrapolated ΔT with different substrates given a constant PD.

and diamond ($\kappa = 2200 \text{ W} \cdot \text{m}^{-1} \cdot \text{K}^{-1}$ [23]) substrates have ΔT reduction by factors of 2.5, 6, and 13, respectively, compared with SiO_2/Si substrate with thick SiO_2 ($\kappa = 1.5 \text{ W} \cdot \text{m}^{-1} \cdot \text{K}^{-1}$ [24]).

Motivated by the thermal studies, an In_2O_3 transistor with T_{ch} of 2.5 nm and L_{ch} of 100 nm is built up on a diamond substrate to alleviate SHE. Fig. 5 presents its dc output and transfer characteristics where an ultrahigh I_D of 3.7 $\text{mA}/\mu\text{m}$ is realized at V_{DS} of 1.4 V without observable SHE even with high PD, in great contrast to Fig. 1(e). The extracted field-effect mobility (μ_{FE}) and SS are 55.6 $\text{cm}^2 \cdot \text{V}^{-1} \cdot \text{s}^{-1}$ and 185 mV/dec, respectively. Note that the degradation of the transfer characteristics from Figs. 1(b) to 5(b) is due to the much thicker T_{ch} as the electrical properties such as V_T ,

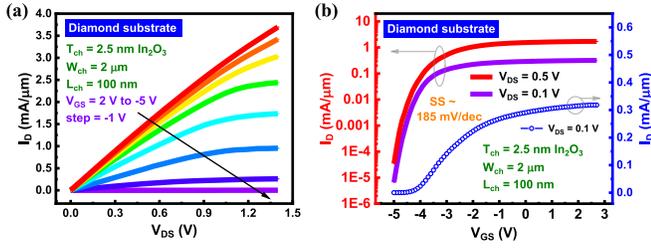


Fig. 5. (a) Output and (b) transfer characteristics of an In_2O_3 FET with L_{ch} of 100 nm and T_{ch} of 2.5 nm on a diamond substrate, performing ultrahigh I_D of 3.7 mA/ μm under dc condition.

ON-OFF ratio, and SS value of ALD In_2O_3 devices are sensitively T_{ch} -dependent [2], [3], [4], [5], [6], [7].

IV. TRANSIENT CHARACTERIZATION

Although diamond with ultrahigh κ is an excellent heat sinker, it is not economically affordable for mass production. TG transistors on Si substrate are much more practically desired. To further mitigate the SHE on a Si substrate, detailed transient thermal characteristics are studied. Fig. 6(a) illustrates the working mechanism for transient thermal characteristics. Similar to steady-state measurements, an active and a passive image are captured by the synchronized CCD camera in each period. The difference is that steady-state investigations only take active images after the steady-state is reached, while transient-state explorations take them throughout the whole process. By setting the active image to be captured at a specific time, the ΔT distribution of that moment can be obtained through the same process shown in Fig. 2(c).

To experimentally observe the time-resolved self-heat-up and cool-down processes, the V_{DS} pulses are set to start at 0 ns and end at 600 ns. Besides, the image capture time window is decreased to 50 ns, which is the minimum of the equipment capability and will be the resolution of the time-resolved measurement. From $t = 0$ –1200 ns, TR images are taken every 50 ns. The heat-up process is illustrated in Fig. 6(b)–6(e) where the V_{DS} pulses begin at $t = 0$ ns. The ΔT distribution around the channel region gradually increases after then due to SHE, and the transient process is observed. Similarly, the V_{DS} pulses end at $t = 600$ ns, and the ΔT peak introduced by SHE gradually decreases as shown in Fig. 6(f)–6(i). The maximum ΔT of each moment of the devices with SiO_2/Si and HR Si substrates are summarized in Fig. 7(a). Two devices with the same structure and dimensions of 400 nm L_{ch} and 2 μm W_{ch} but different substrates are measured. It takes roughly 350 and 300 ns to reach steady-states for heat-up and cool-down, no matter whether SiO_2/Si or HR Si serves as the substrate.

By assuming the processes satisfy Newton's law of cooling [25] (the rate of heat transfer is proportional to the temperature difference between the body and its surroundings) and utilizing their individual initial conditions, the following equations can be derived [26]:

$$\Delta T(t) = \Delta T_{\text{st}} \times (1 - e^{-t/\tau})$$

for heating-up and

$$\Delta T(t) = \Delta T_{\text{st}} \times e^{-t/\tau}$$

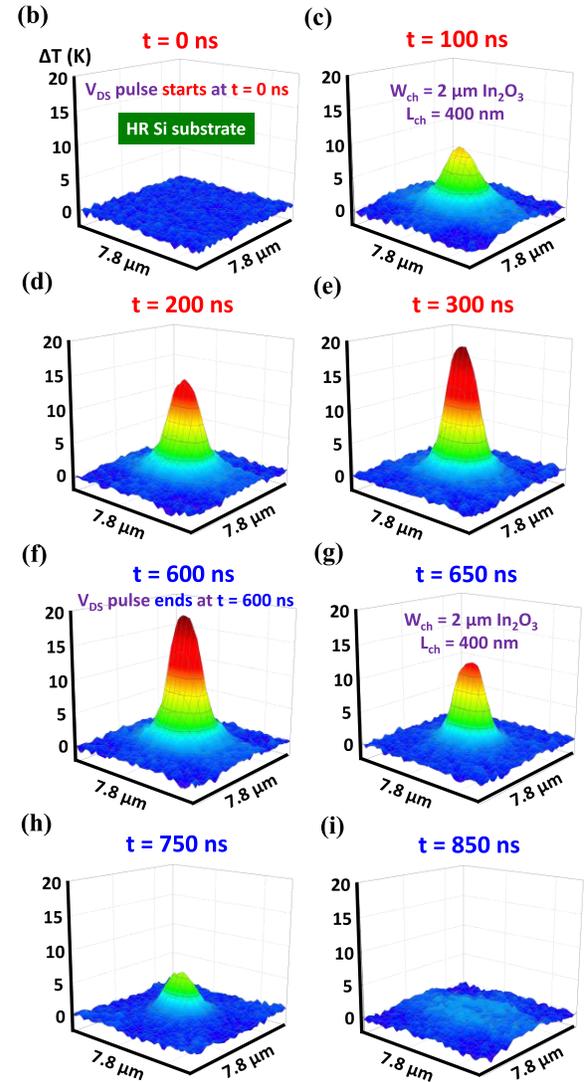
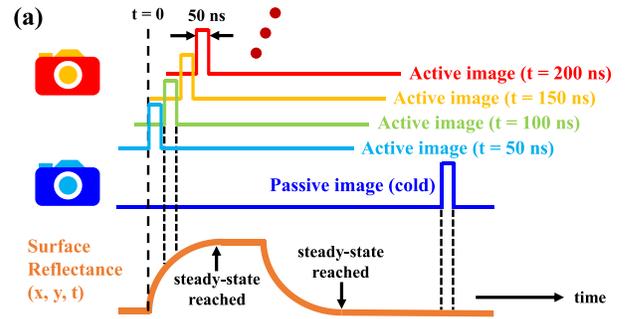


Fig. 6. (a) Working mechanism of transient thermal property study with an ultrafast HR TR imaging system. (b)–(i) Transient TR characteristics of a TG ALD In_2O_3 FET on HR Si substrate. The device was self-heated-up by applying a V_{DS} pulse starting at 0 ns and ending at 600 ns. (b)–(e) Demonstrate the heat-up process while (f)–(i) demonstrate the cool-down process.

for cooling down where $\Delta T(t)$ represents the ΔT at time t , ΔT_{st} represents the steady-state ΔT , and τ represents the thermal time constant. Accordingly, the following equations are obtained:

$$\frac{t}{\tau_h} = -\ln\left(1 - \frac{\Delta T(t)}{\Delta T_{\text{st}}}\right)$$

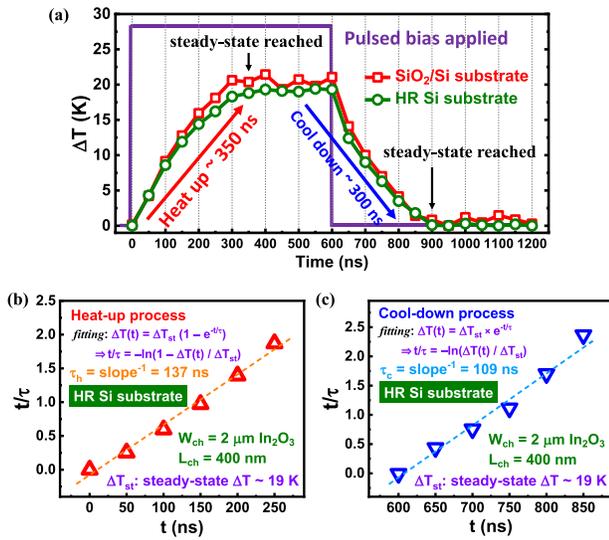


Fig. 7. (a) Transient ΔT results and comparison with different substrates. Roughly 325 ns is needed to reach steady-state in both cases. Time constant extraction of (b) heat-up and (c) cool-down processes from the transient ΔT measurement results with HR Si substrate of Fig. 7(a). The t/τ values are obtained by plugging the measured individual $\Delta T(t)$ values into the indicated formulas.

for heating-up and

$$\frac{t}{\tau_c} = -\ln\left(\frac{\Delta T(t)}{\Delta T_{st}}\right)$$

for cooling-down where τ_h and τ_c represent the heat-up and cool-down thermal time constant, respectively. Fig. 7(b) and (c) demonstrates the calculated t/τ values by plugging the experimentally observed data points in Fig. 7(a) at time t into the dedicated equations. The heat-up and cool-down thermal time constants, τ_h and τ_c , can be consequently acquired by extracting the inversed slopes of the regression lines in Fig. 7(b) and (c), respectively. Finally, τ_h and τ_c are extracted to be 137 and 109 ns, respectively.

With the understanding of its transient thermal characteristics, short-pulse electrical co-optimization is proposed to alleviate SHE. Fig. 8(a) exhibits the working principles of pulse measurement in the time domain where both the drain and gate biases are pulses instead of constant. At the start of each pulse, t_{rise} of time is used for setting the biases to the desired values. Then, they last for t_{pulse} of time in which the first t_{delay} of time is used for bias stabilization, and the following t_{meas} of time is used for current measurement. The t_{delay} time is required as the biases show some damping after being changed. After t_{pulse} , t_{fall} of time is used for setting the biases back to zero, and they stay zero until the start of the next cycle. By applying pulse measurement, the device under test is only turned on for t_{pulse} of time each cycle. As long as t_{pulse} is short enough, the device under test is turned off before SHE degrades the device's performance. Hence, the SHE can be mostly avoided.

Fig. 8(b) exhibits the I_D - V_{DS} curves of a TG In₂O₃ FET with T_{ch} of 1.9 nm and L_{ch} of 80 nm on the HR Si substrate. The empty symbols represent the dc measurement results where V_{DS} up to 1.8 V can be applied to achieve I_D up to

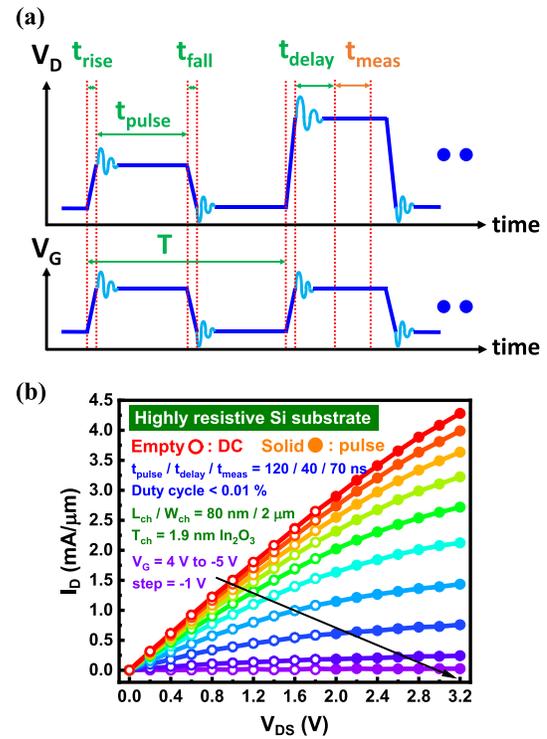


Fig. 8. (a) Diagram sketch of pulse measurement setup in time domain. The light blue damping illustrates the bias stabilization process after being set to desired values. (b) Output characteristics of a TG In₂O₃ FET with short L_{ch} of 80 nm and T_{ch} of 1.9 nm on HR Si substrate achieving extremely high I_D up to 4.3 mA/ μm with short-pulse measurement.

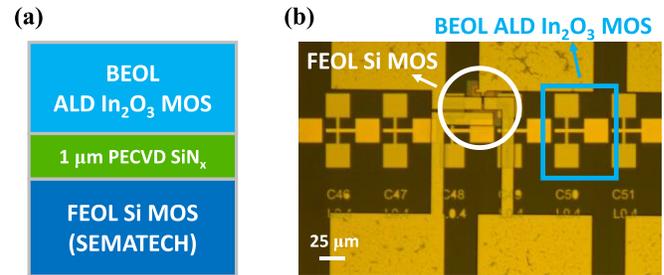


Fig. 9. (a) Structure illustration of combining BEOL ALD In₂O₃ transistors and FEOL Si devices with 1- μm -thick PECVD SiN_x. (b) Optical photograph of the structure in (a) where the indicated FEOL and BEOL devices are vertically separated by a 1- μm -thick PECVD SiN_x layer.

2.6 mA/ μm without observable SHE in its electrical performance. Nonetheless, SHE starts to take place with larger V_{DS} with dc biases. On the other hand, the solid symbols represent the pulse-measurement results with t_{pulse} of 120 ns. With short-pulse measurement, an ultrahigh I_D of 4.3 mA/ μm is achieved at a higher V_{DS} of 3.2 V. In great contrast to Fig. 1(d), even with such high PD on an atomically thin channel, there is no observable SHE since the t_{pulse} of 120 ns is even shorter than a heat-up thermal time constant, τ_h , of 137 ns. The key fact is that the electrical response is much faster than the thermal response. Short-pulse electrical measurement can significantly reduce SHE and probe the potential of the material and device performance.

Besides, SiN_x is able to serve as a promising interlayer dielectric for M3D integration due to its relatively high

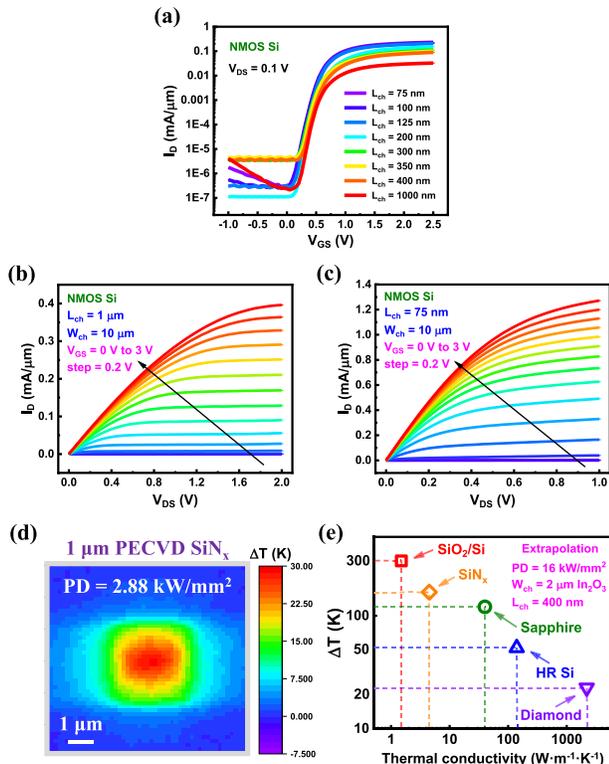


Fig. 10. (a) Transfer characteristics of the FEOL Si nMOS, showing decent switching behaviors and E-mode operation. (b) Corresponding output characteristics of a long channel device with L_{ch} of $1 \mu\text{m}$, performing great saturation behaviors. (c) Similar I_D - V_{DS} curves of a short channel device with L_{ch} of 75 nm , exhibiting high I_D of $1.2 \text{ mA}/\mu\text{m}$ at V_{DS} of 1 V . (d) ΔT distribution of a TG In_2O_3 transistor with channel thickness of 1.8 nm shown in Fig. 9(b). (e) Comparison of the extrapolated ΔT with different substrates including PECVD SiN_x given a constant PD.

thermal conductivity ($4.5 \text{ W}\cdot\text{m}^{-1}\cdot\text{K}^{-1}$ [27]), low leakage current [28], [29], great CMOS compatibility [30], and low growth temperature enabled by PECVD [31], [32]. To integrate BEOL-compatible TG ALD In_2O_3 TFTs on the FEOL devices, a $1\text{-}\mu\text{m}$ -thick layer of SiN_x was grown by PECVD at a low temperature of $150 \text{ }^\circ\text{C}$ as an interlayer on SEMATECH chips with Si transistors to maintain decent heat dissipation capability. TG ALD In_2O_3 transistors were then fabricated on top of the thick SiN_x interlayer as illustrated in the diagram sketch of Fig. 9(a). Fig. 9(b) exhibits an optical image of the fabricated devices where a FEOL Si device and a BEOL ALD In_2O_3 device are indicated in the white circle and blue square, respectively. The FEOL and BEOL transistors are separated by a $1\text{-}\mu\text{m}$ -thick SiN_x layer in between for satisfactory electrical insulation.

Fig. 10(a) shows the transfer characteristics of the SEMATECH E-mode Si nMOS with variant L_{ch} from 75 nm to $1 \mu\text{m}$ at V_{DS} of 0.1 V where great switching behaviors are exhibited regardless of L_{ch} . ON-OFF ratios range from 4 to 6 orders of magnitude, and SS values are all approximately $90\text{--}110 \text{ mV}/\text{dec}$. Fig. 10(b) illustrates the corresponding output characteristics of a long channel device with L_{ch} of $1 \mu\text{m}$ where exceptional saturation properties are exhibited. Besides, Fig. 10(c) similarly reveals the I_D - V_{DS} curves of a device with a short L_{ch} of 75 nm which reaches a high I_D of $1.2 \text{ mA}/\mu\text{m}$ at V_{DS} of 1 V . The electrical characterization of the FEOL Si devices was measured before the deposition of the SiN_x layer

and the fabrication of the BEOL devices. Fig. 10(d) shows the TR measurement of a TG In_2O_3 transistor with L_{ch} of 400 nm , W_{ch} of $2 \mu\text{m}$, and channel thickness of 1.8 nm in the FEOL- SiN_x -BEOL structure. It is noticeable that the hottest region performs roughly wider ΔT distribution compared with previous cases shown in Fig. 3, which might be due to the reflected heat from the SiN_x /FEOL interface to the surface with the presence of the $1\text{-}\mu\text{m}$ -thick SiN_x layer. Fig. 10(e) illustrates the comparison of the thermal dissipation capability of the devices on PECVD SiN_x with that on other substrates, indicating that the PD-normalized ΔT of In_2O_3 transistors in the FEOL- SiN_x -BEOL structure is approximately 1.9 times lower than that on a SiO_2/Si substrate.

Nevertheless, the TG ALD In_2O_3 transistors fabricated on the SiN_x -FEOL structure demonstrate poor switching behaviors with an ON-OFF ratio smaller than 1 order of magnitude. This is plausibly due to the defect-rich surface of the PECVD SiN_x layer [33], [34], [35]. More investigations and efforts are needed to better realize the integration of FEOL and BEOL fabrications, but this work proposed the FEOL-interlayer-BEOL structure where the interlayer not only separates the FEOL and BEOL devices but also serves as a thermally conductive layer to alleviate the SHE of the BEOL ALD In_2O_3 transistors. The realization of this structure also benefits the applications of ALD In_2O_3 transistors toward M3D integrations at advanced technology nodes.

V. CONCLUSION

In summary, transient thermal and electrical properties of TG ALD In_2O_3 FETs on various thermally conductive substrates are co-optimized employing an ultrafast HR TR imaging technique to unveil the problematic SHE. By using HR Si substrate and electrical short-pulse measurement, an ultrahigh I_D of $4.3 \text{ mA}/\mu\text{m}$ is achieved on atomically thin In_2O_3 devices without observable SHE. This work demonstrates that the understanding of both thermal and electrical transient dynamics is important to resolve thermal bottleneck on atomically thin oxide semiconductor devices. Besides, a FEOL-interlayer-BEOL structure is proposed where the interlayer not only electrically isolates the FEOL and BEOL devices but also serves as a thermally conductive layer to alleviate the SHE and exhibit the potentials for the applications of BEOL ALD In_2O_3 TFTs toward M3D integration.

REFERENCES

- [1] K. Nomura, A. Takagi, T. Kamiya, H. Ohta, M. Hirano, and H. Hosono, "Amorphous oxide semiconductors for high-performance flexible thin-film transistors," *Jpn. J. Appl. Phys.*, vol. 45, no. 5, pp. 4303-4308, Apr. 2006, doi: 10.1143/JJAP.45.4303.
- [2] M. Si et al., "Why In_2O_3 can make 0.7 nm atomic layer thin transistors," *Nano Lett.*, vol. 21, no. 4, pp. 500-506, Jan. 2021, doi: 10.1021/acs.nanolett.0c03967.
- [3] M. Si, Z. Lin, A. Charnas, and P. D. Ye, "Scaled atomic-layer-deposited indium oxide nanometer transistors with maximum drain current exceeding $2 \text{ A}/\text{mm}$ at drain voltage of 0.7 V ," *IEEE Electron Device Lett.*, vol. 42, no. 2, pp. 184-187, Feb. 2021, doi: 10.1109/LED.2020.3043430.
- [4] M. Si, A. Charnas, Z. Lin, and P. D. Ye, "Enhancement-mode atomic-layer-deposited In_2O_3 transistors with maximum drain current of $2.2 \text{ A}/\text{mm}$ at drain voltage of 0.7 V by low-temperature annealing and stability in hydrogen environment," *IEEE Trans. Electron Devices*, vol. 68, no. 3, pp. 1075-1080, Mar. 2021, doi: 10.1109/TED.2021.3053229.

- [5] M. Si, Z. Lin, Z. Chen, and P. D. Ye, "High-performance atomic-layer-deposited indium oxide 3-D transistors and integrated circuits for monolithic 3-D integration," *IEEE Trans. Electron Devices*, vol. 68, no. 12, pp. 6605–6609, Dec. 2021, doi: [10.1109/TED.2021.3106282](https://doi.org/10.1109/TED.2021.3106282).
- [6] M. Si, Z. Lin, Z. Chen, X. Sun, H. Wang, and P. D. Ye, "Scaled indium oxide transistors fabricated using atomic layer deposition," *Nature Electron.*, vol. 5, no. 3, pp. 164–170, Feb. 2022, doi: [10.1038/s41928-022-00718-w](https://doi.org/10.1038/s41928-022-00718-w).
- [7] P.-Y. Liao, M. Si, Z. Zhang, Z. Lin, and P. D. Ye, "Realization of maximum 2 A/mm drain current on top-gate atomic-layer-thin indium oxide transistors by thermal engineering," *IEEE Trans. Electron Devices*, vol. 69, no. 1, pp. 147–151, Jan. 2022, doi: [10.1109/TED.2021.3125923](https://doi.org/10.1109/TED.2021.3125923).
- [8] P.-Y. Liao et al., "Thermal studies of BEOL-compatible top-gated atomically thin ALD In₂O₃ FETs," in *Proc. IEEE Symp. VLSI Technol.*, Jun. 2022, pp. 322–323.
- [9] J. Wu, F. Mo, T. Saraya, T. Hiramoto, and M. Kobayashi, "A monolithic 3D integration of RRAM array with oxide semiconductor FET for in-memory computing in quantized neural network AI applications," in *Proc. IEEE Symp. VLSI Technol.*, Jun. 2020, pp. 1–2.
- [10] W. Chakraborty, B. Grisafe, H. Ye, I. Lightcap, K. Ni, and S. Datta, "BEOL compatible dual-gate ultra thin-body W-doped indium-oxide transistor with $I_{on} = 370 \mu A/\mu m$, $SS = 73 \text{ mV/dec}$ and I_{on}/I_{off} ratio $> 4 \times 10^9$," in *Proc. IEEE Symp. VLSI Technol.*, Jun. 2020, pp. 1–2.
- [11] S. Samanta, K. Han, C. Sun, C. Wang, A. V.-Y. Thean, and X. Gong, "Amorphous IGZO TFTs featuring extremely-scaled channel thickness and 38 nm channel length: Achieving record high $G_{m,max}$ of $125 \mu S/\mu m$ at V_{DS} of 1 V and I_{ON} of $350 \mu A/\mu m$," in *Proc. IEEE Symp. VLSI Technol.*, Jun. 2020, pp. 1–2.
- [12] S. Li et al., "Nanometre-thin indium tin oxide for advanced high-performance electronics," *Nature Mater.*, vol. 18, no. 10, pp. 1091–1097, Oct. 2019, doi: [10.1038/s41563-019-0455-8](https://doi.org/10.1038/s41563-019-0455-8).
- [13] S. Subhechha et al., "Ultra-low leakage IGZO-TFTs with raised source/drain for $V_t > 0 \text{ V}$ and $I_{on} > 30 \mu A/\mu m$," in *Proc. IEEE Symp. VLSI Technol.*, Jun. 2022, pp. 292–293.
- [14] A. Charnas, M. Si, Z. Lin, and P. D. Ye, "Realization of enhancement-mode atomic-layer thin In₂O₃ transistors with maximum current exceeding 2 A/mm at drain voltage of 0.7 V enabled by oxygen plasma treatment," *Appl. Phys. Lett.*, vol. 118, no. 5, Feb. 2021, Art. no. 052107, doi: [10.1063/5.0039783](https://doi.org/10.1063/5.0039783).
- [15] U. Chand et al., "Sub-10 nm ultra-thin ZnO channel FET with record-high $561 \mu A/\mu m$ I_{ON} at V_{DS} 1V, high μ -84 $\text{cm}^2/\text{V}\cdot\text{s}$ and 1T-1RRAM memory cell demonstration memory implications for energy-efficient deep-learning computing," in *Proc. IEEE Symp. VLSI Technol.*, Jun. 2022, pp. 326–327.
- [16] J. Zhang et al., "Fluorine-passivated In₂O₃ thin film transistors with improved electrical performance via low-temperature CF₄/N₂O plasma," *Appl. Phys. Lett.*, vol. 121, no. 17, Oct. 2022, Art. no. 172101, doi: [10.1063/5.0113015](https://doi.org/10.1063/5.0113015).
- [17] Z. Zhang et al., "A gate-all-around In₂O₃ nanoribbon FET with near 20 mA/ μm drain current," *IEEE Electron Device Lett.*, vol. 43, no. 11, pp. 1905–1908, Nov. 2022, doi: [10.1109/LED.2022.3210005](https://doi.org/10.1109/LED.2022.3210005).
- [18] K. Maize, A. Ziabari, W. D. French, P. Lindorfer, B. O'Connell, and A. Shakouri, "Thermoreflectance CCD imaging of self-heating in power MOSFET arrays," *IEEE Trans. Electron Devices*, vol. 61, no. 9, pp. 3047–3053, Sep. 2014, doi: [10.1109/TED.2014.2332466](https://doi.org/10.1109/TED.2014.2332466).
- [19] H. Zhou, K. Maize, J. Noh, A. Shakouri, and P. D. Ye, "Thermodynamic studies of β -Ga₂O₃ nanomembrane field-effect transistors on a sapphire substrate," *ACS Omega*, vol. 2, no. 11, pp. 7723–7729, Nov. 2017, doi: [10.1021/acsomega.7b01313](https://doi.org/10.1021/acsomega.7b01313).
- [20] J. Noh et al., "Enhancement of thermal transfer from β -Ga₂O₃ nanomembrane field-effect transistors to high thermal conductivity substrate by inserting an interlayer," *IEEE Trans. Electron Devices*, vol. 69, no. 3, pp. 1186–1190, Jan. 2021, doi: [10.1109/TED.2022.3142651](https://doi.org/10.1109/TED.2022.3142651).
- [21] S. Burghartz and B. Schulz, "Thermophysical properties of sapphire, AlN and MgAl₂O₃ down to 70 K," *J. Nucl. Mater.*, vol. 212, pp. 1065–1068, Sep. 1994, doi: [10.1016/0022-3115\(94\)90996-2](https://doi.org/10.1016/0022-3115(94)90996-2).
- [22] H. R. Shanks, P. D. Maycock, P. H. Sidles, and G. C. Danielson, "Thermal conductivity of silicon from 300 to 1400 K," *Phys. Rev.*, vol. 130, pp. 1743–1748, Jun. 1963, doi: [10.1103/PhysRev.130.1743](https://doi.org/10.1103/PhysRev.130.1743).
- [23] J. R. Olson, R. O. Pohl, J. W. Vandersande, A. Zoltan, T. R. Anthony, and E. F. Banholzer, "Thermal conductivity of diamond between 170 and 1200 K and the isotope effect," *Phys. Rev. B, Condens. Matter*, vol. 47, no. 22, pp. 14850–14857, Jun. 1993, doi: [10.1103/PhysRevB.47.14850](https://doi.org/10.1103/PhysRevB.47.14850).
- [24] H.-C. Chien, D.-J. Yao, M.-J. Huang, and T.-Y. Chang, "Thermal conductivity measurement and interface thermal resistance estimation using SiO₂ thin film," *Rev. Sci. Instrum.*, vol. 79, no. 5, May 2008, Art. no. 054902, doi: [10.1063/1.2927253](https://doi.org/10.1063/1.2927253).
- [25] I. Newton, "Scala graduum caloribus. Calorum descriptiones & signa," *Philos. Trans. Roy. Soc.*, vol. 22, pp. 824–829, 1701.
- [26] R. W. Lewis, P. Nithiarasu, and K. N. Seetharamu, *Fundamentals of the Finite Element Method for Heat and Fluid Flow*. Hoboken, NJ, USA: Wiley, 2004.
- [27] P. Eriksson, J. Y. Andersson, and G. Stemme, "Thermal characterization of surface-micromachined silicon nitride membranes for thermal infrared detectors," *J. Microelectromech. Syst.*, vol. 6, no. 1, pp. 55–61, Mar. 1997, doi: [10.1109/84.557531](https://doi.org/10.1109/84.557531).
- [28] M. Hua et al., "Characterization of leakage and reliability of SiN_x gate dielectric by low-pressure chemical vapor deposition for GaN-based MIS-HEMTs," *IEEE Trans. Electron Devices*, vol. 62, no. 10, pp. 3215–3222, Sep. 2015, doi: [10.1109/TED.2015.2469716](https://doi.org/10.1109/TED.2015.2469716).
- [29] K. Sekine, Y. Saito, M. Hirayama, and T. Ohmi, "Highly robust ultrathin silicon nitride films grown at low-temperature by microwave-excitation high-density plasma for giga scale integration," *IEEE Trans. Electron Devices*, vol. 47, no. 7, pp. 1370–1374, Jul. 2000, doi: [10.1109/16.848279](https://doi.org/10.1109/16.848279).
- [30] A. Frigg et al., "Low loss CMOS-compatible silicon nitride photonics utilizing reactive sputtered thin films," *Opt. Exp.*, vol. 27, pp. 37795–37805, Dec. 2019, doi: [10.1364/OE.380758](https://doi.org/10.1364/OE.380758).
- [31] H.-Y. Chang, C.-Y. Meng, C.-W. Huang, and S.-C. Lee, "The low-temperature a -SiN_x films with high impermeability and high optical gap with application to organic light-emitting diode," *J. Appl. Phys.*, vol. 98, no. 8, Oct. 2005, Art. no. 084501, doi: [10.1063/1.2089161](https://doi.org/10.1063/1.2089161).
- [32] J. R. Lothian, F. Ren, S. J. Pearton, C. R. Abernathy, B. Tseng, and W. S. Hobson, "Low temperature SiN_x as a sacrificial layer in novel device fabrication," *MRS Proc.*, vol. 300, pp. 161–167, Dec. 1993, doi: [10.1557/PROC-300-161](https://doi.org/10.1557/PROC-300-161).
- [33] S.-J. Tsai et al., "Approaching defect-free amorphous silicon nitride by plasma-assisted atomic beam deposition for high performance gate dielectric," *Sci. Rep.*, vol. 6, pp. 1–9, Jun. 2016, doi: [10.1038/srep28326](https://doi.org/10.1038/srep28326).
- [34] C. Zhang, M. Wu, P. Wang, M. Jian, J. Zhang, and L. Yang, "Stability of SiN_x prepared by plasma-enhanced chemical vapor deposition at low temperature," *Nanomaterials*, vol. 11, no. 12, p. 3363, Dec. 2021, doi: [10.3390/nano11123363](https://doi.org/10.3390/nano11123363).
- [35] J. Kanicki and W. L. Warren, "Defects in amorphous hydrogenated silicon nitride films," *J. Non. Cryst. Solids*, vol. 164, pp. 1055–1060, Dec. 1993, doi: [10.1016/0022-3093\(93\)91180-B](https://doi.org/10.1016/0022-3093(93)91180-B).