

Gas Sensing Properties of P-type Semiconducting Cr-doped TiO₂ Thin Films

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SUMMARY

Cr₂O₃-TiO₂ thin films have been prepared by the sol-gel process. Titanium butoxide was used as the precursor material. The solution was mixed with a chromium compound then spun onto sapphire and silicon substrates at 2500 rpm for 30 s. The films were annealed at temperatures of between 400 and 700°C for 1 hr. The X-Ray Diffraction (XRD), Scanning Electronic Microscope (SEM), Rutherford Backscatter Spectrometry (RBS) and X-ray Photoelectron Spectroscopy (XPS) techniques were employed for microstructural characterizations. The response to both NO₂ and O₂ gas confirmed that the films are of a p-type behaviour between 350 and 400°C. The films showed a good response to oxygen, in the range from 100 ppm to 10% of O₂ at an operating temperature of 370°C. The response is also fast and stable. The p-type Cr-doped TiO₂ thin films have potential for development of a novel gas sensors.

Keywords: Cr-doped TiO₂, P-type semiconductor, Sol-gel process.

INTRODUCTION

TiO₂ applications ranging from catalytic and electrochemical processes through optical coatings to gas sensing devices are rapidly expanding. Consequently, new and important issues continue to arise. A major concern is the effect of doping the electronic structure of TiO₂ and its impact on its gas sensing performance. A platinum doped titania film oxygen sensor integrated with a temperature compensating thermistor was fabricated by means of tripole magnetron sputtering. This sensor has a fast and stable response [1]. The influence of donor type (Nb⁵⁺), acceptor type (Cr³⁺) or Sn⁴⁺ additions isovalent with Ti⁴⁺

on the electronic structure of r.f.-sputtered TiO₂ thin films and its subsequent effect on its gas sensing characteristics were reported [2, 3]. The effect of iron doping on TiO₂ thin films showed that the iron causes a structural transformation from anatase to rutile. The electrical measurements indicate that iron acts as an acceptor impurity [4]. The gas sensing properties of the binary metal oxide TiO₂-WO₃, TiO₂-MoO₃, TiO₂-V₂O₅ have also been studied extensively [5, 6].

Most metal oxide thin films are n-type semiconducting materials. The purpose of the present study is to enhance the performance of the gas sensing thin films, and most importantly to develop a novel p-type semiconducting gas sensing film. Different mole ratios of 10%-90% Cr doped in TiO₂ thin films were prepared by the sol-gel process. The structure and electrical properties of the Cr₂O₃-TiO₂ thin films are reported in this paper.

EXPERIMENTAL

Preparation of Cr-doped TiO₂ Thin Films

The precursor solution for the sol-gel process was prepared from titanium butoxide and dissolved in butanol. The spin-coating technique was employed (at a speed of 2500 rpm for 30 sec) to deposit the films onto sapphire substrates with Pt-film interdigital electrodes on the front-side and a Pt-film heater on the backside. Silicon substrates were used for microstructure characterization. The as-deposited films were left open in air for 12 hours and subsequently annealed at various temperatures, viz. 400°C, 500°C, 600°C and 700°C, for 1 hr.

Microstructure Characterization

The microstructure and the surface topography of the films were examined using an SEM (Philips XL-30) operating at 30 kV. The crystal structure of the films was studied using a Bruker D8 Advanced XRD Diffractometer operating at 40 mA and 40 kV, Cu-K α_1 radiation ($\lambda=0.154$ nm) mono-chromated with a graphite sample monochromator. The RBS analysis was performed using a 2.0 MeV He $^{2+}$ ion beam (1.5 mm diameter) accelerated by a Van de Graaf accelerator. The detector was fixed at 169° to the beam direction. Spectra were accumulated up to a total charge of 20 μ C. The chemical composition of the thin films was examined using an XPS on a VG Microlab 310F.

Gas Sensing Measurements

The gas sensing properties of the films to O $_2$ were measured using a computerized multimeter system (34401A Hewlet-Packard) and a gas calibration system incorporating mass flow controllers set at 0.2 LPM. The gas response, S , is defined as $S = R_g/R_b$, where R_g is the electrical resistance at different O $_2$ concentrations and R_b is the resistance at baseline (10% ppm O $_2$). The measurements were carried out at different operating temperatures between 220 and 420°C. The ambient temperature was 20°C and RH 30%. Certified O $_2$ gas cylinders of 100 ppm, 1000 ppm, 1% and 10% balanced with dry N $_2$ were used. All results presented are referenced to samples annealed at 600°C unless otherwise stated.

RESULTS AND DISCUSSION

Microstructure Characterizations

Fig. 1(a) shows a secondary electron image of Cr (20%)-doped TiO $_2$ and Fig. 1(b) shows the Backscatter Electron (BSE) image of the film on silicon substrate annealed at 600°C for 1 hr. There are two different regions, black circle-like and grey coloured areas. From the BSE, we may conclude that the black circle-like areas are mainly TiO $_2$ dominated and the other areas are Cr $_2$ Ti $_2$ O $_7$ phase (as confirmed by XRD).

The incorporation of Cr in TiO $_2$ lattice does not affect the crystallography of the pure TiO $_2$ material. It is known that TiO $_2$ films crystallise in the rutile phase during high temperature annealing ($T>600^\circ\text{C}$) in an oxidizing atmosphere.

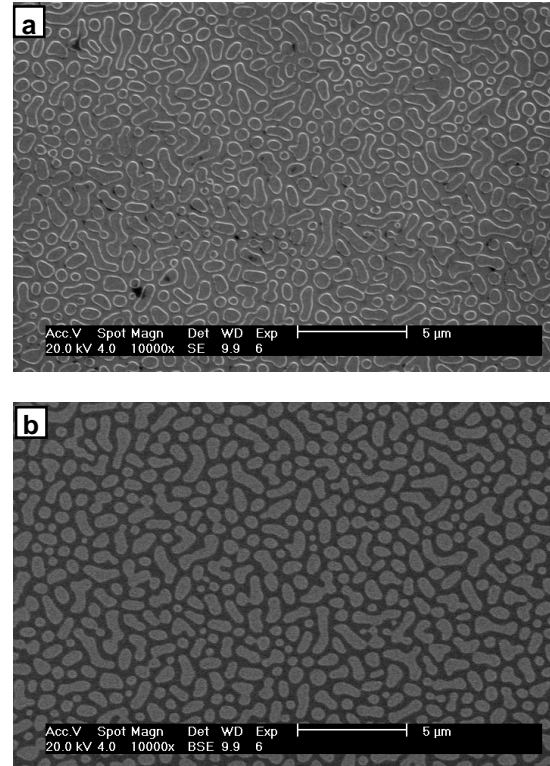


Fig. 1: SEM images of Cr-doped TiO $_2$ thin film annealed at 600°C on Si substrate. (a) secondary electron (SE) and (b) backscatter electron (BSE).

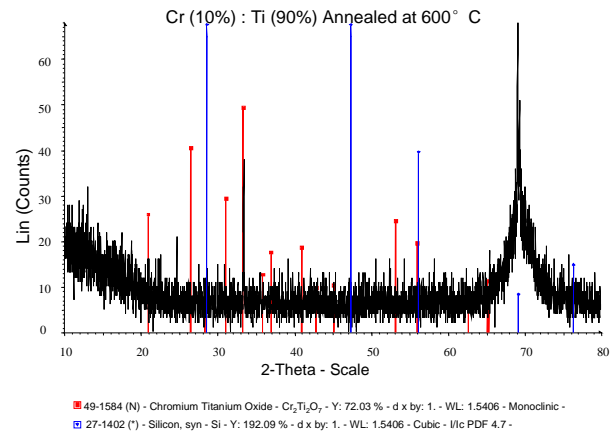


Fig. 2: XRD pattern of Cr-TiO $_2$ film annealed at 600°C on a Si substrate.

It has been shown that no secondary phases resulting from Cr and Nb doping up to 4 at.% Cr and 6 at.% Nb are observed in XRD pattern [3]. From the XRD results, it is believed that this is also the case for the sol-gel prepared Cr-doped TiO $_2$ with 5 at. % Cr. However, as seen from Fig. 2, (the thin film of 10 at. % Cr

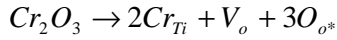
annealed at 600°C) a weak reflection of $\text{Cr}_2\text{Ti}_2\text{O}_7$ at 2.71 \AA and a Si peak at approximately 1.35 \AA were observed.

Based on the RBS results of the Cr-doped TiO_2 thin film, it is found that the thickness of the films is 20 nm. As the resolution limit of the detectors used for RBS gives a lower limit of approximately 80nm-100nm, the depth profile information could not be determined for the TiO_2 films. The chromium and titanium peaks shown in Fig.3 are very small and not resolvable due to the fact that the atomic weights are too close. The films have a high purity without any other metal contamination.

Fig. 4 shows XPS scan spectra of the Cr-doped TiO_2 thin film. The Ti_{2p} and Cr_{2p} spectra indicates the chemical states of Ti and Cr to be Ti^{4+} and Cr^{3+} , respectively. Two O_{1s} peaks were observed corresponding to O^{2-} belonging to Ti or Cr and to silicon (SiO_2). Similarly, the Si_{2p} spectra indicated Si-Si bonding (substrate) and SiO_2 due to oxide formation on the silicon surface. XPS also confirmed the expected atomic concentrations of the films as shown in Table 1.

Gas Sensing Properties

The trivalent Cr^{+3} acts as an acceptor type impurity which can be expressed as:



where V_o represents oxygen vacancies and Cr_{Ti} is Cr substitution in Ti sites. The semiconductor sensing mechanism is based on reactions between the surface of the film and gas molecules, causing a change in the semiconductor's resistance due to a charge transfer. For an n-type semiconductor, the resistance increases, due to the electron capture by the oxidising gas; for p-type, an increase in conductance is found.

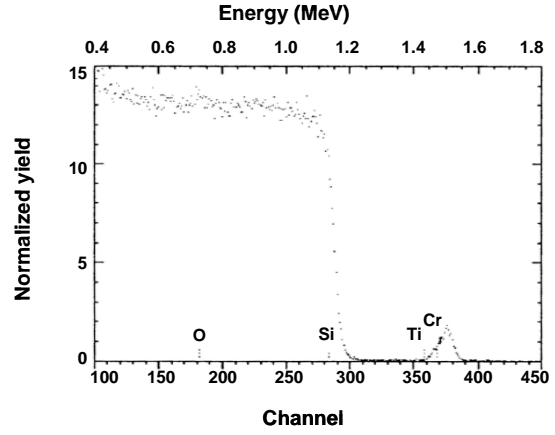


Fig. 3: RBS spectra of Cr- TiO_2 (50%) thin film.

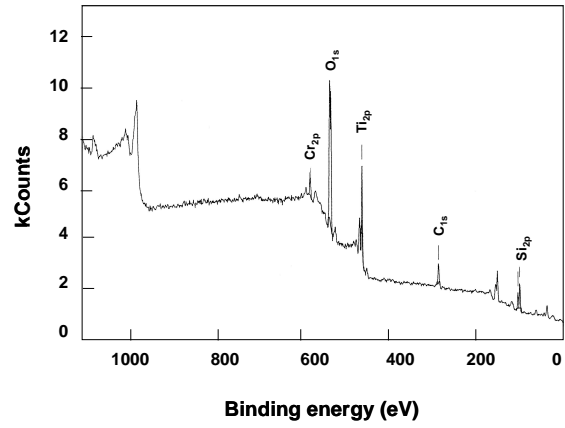


Fig.4: XPS scan spectra of the Cr-doped TiO_2 thin film annealed at 600°C on a Si substrate.

Table 1: XPS data for Cr-doped TiO_2 thin films.

Sample	Cr 2p (Cr^{3+})	Ti 2p (Ti^{4+})	O 1s (SiO_2+OH^-)	O 1s (O^{2-})	Si 2p (SiO_2)	Si 2p (Si-Si)	Ti/Cr Ratio
1	1.453	9.158	38.285	21.860	9.221	20.023	6.3
2	2.046	8.295	32.786	26.491	13.542	16.840	4.1
3	1.897	7.670	37.232	25.062	15.960	12.179	4.0
4	1.412	6.579	39.844	20.992	17.702	13.471	4.7
5	1.510	6.660	39.774	21.625	17.107	13.324	4.4
6	2.414	9.258	32.362	29.932	17.143	8.891	3.8
7	2.904	11.062	28.494	37.611	9.369	10.560	3.8

Fig. 5 shows the dynamic responses of 35 at.% Cr doped TiO₂ thin film to 100 ppm, 1000 ppm, 1% and 10% oxygen at an operating temperature of 370°C. The baseline is at an oxygen concentration of 10%. The response is fast, stable and repeatable. The incorporation of chromium cations into the TiO₂ lattice initially resulted in an increase in resistivity of the thin films. However, the acceptor behavior of Cr dominated at high partial pressure of oxygen resulting in a change from n-type to p-type conductivity when the dopant concentration increases. The response of a p-type film is clearly opposite to that of n-type, as seen in Fig.5. The response (τ_{res}) and recovery (τ_{rec}) times are parameters qualifying the dynamic performance of the sensor. As shown in Fig.5, the response time is about 1-3 minutes and recovery time is less than 1 minute.

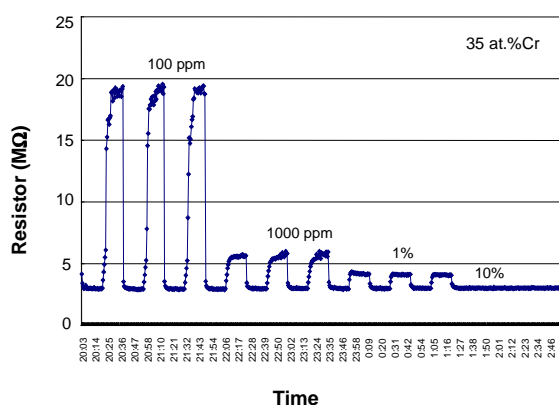


Fig. 5: The response of 35% Cr-doped TiO₂ film to 100 ppm, 1000 ppm and 1% O₂ at an operating temperature of 370°C.

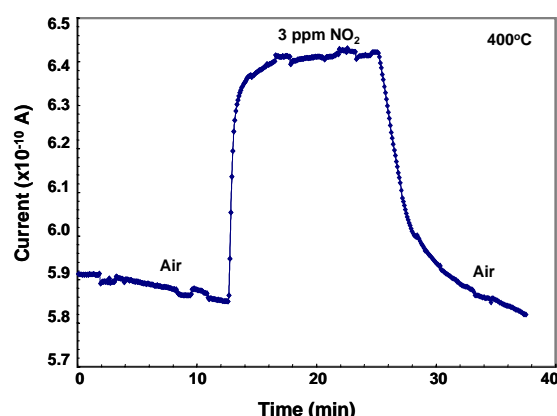


Fig. 6: The response of Cr-doped TiO₂ film to 3 ppm NO₂ at an operating temperature of 400°C.

The films were also exposed to 3 ppm of NO₂ at an operating temperature of 400°C as shown in Fig.6. The films conductivity (when biased at 1 V DC) increases

when NO₂ gas is introduced. The film also exhibits p-type semiconductor behaviour. The response of the film, S is 1.1, the response time τ_{res} is 85 s and recovery time τ_{rec} is 160 s.

CONCLUSIONS

P-type Cr-doped TiO₂ thin films have been successfully prepared by the sol-gel process. The response to both NO₂ and O₂ gas confirmed that the films are of a p-type behaviour between 350 and 400°C. The films showed a fast and stable response to oxygen, in the range from 100 ppm to 10% of O₂ at an operating temperature of 370°C. The p-type Cr-doped TiO₂ thin films have potential for development of a novel gas sensors.

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